2D Materials



Evidence of Spin Frustration in a Vanadium Diselenide Monolayer Magnet

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Monolayer VSe₂, featuring both charge density wave and magnetism phenomena, represents a unique van der Waals magnet in the family of metallic 2D transition-metal dichalcogenides (2D-TMDs). Herein, by means of in situ microscopy and spectroscopic techniques, including scanning tunneling microscopy/ spectroscopy, synchrotron X-ray and angle-resolved photoemission, and X-ray absorption, direct spectroscopic signatures are established, that identify the metallic 1T-phase and vanadium 3d¹ electronic configuration in monolayer VSe₂ grown on graphite by molecular-beam epitaxy. Element-specific X-ray magnetic circular dichroism, complemented with magnetic susceptibility measurements, further reveals monolayer VSe₂ as a frustrated magnet, with its spins exhibiting subtle correlations, albeit in the absence of a long-range magnetic order down to 2 K and up to a 7 T magnetic field. This observation is attributed to the relative stability of the ferromagnetic and antiferromagnetic ground states, arising from its atomic-scale structural features, such as rotational disorders and edges. The results of this study extend the current understanding of metallic 2D-TMDs in the search for exotic low-dimensional quantum phenomena, and stimulate further theoretical and experimental studies on van der Waals monolayer magnets.

Endowed by the many possible combinations of their constituting elements, 2D transition-metal dichalcogenides (2D-TMDs) can exhibit a multitude of exotic properties.^[1,2] For the much-studied group VI 2D semiconductors, these include coupled spin and valley degrees of freedom,^[3] an indirect-to-direct bandgap transition,^[4,5] valley polarization,^[6,7] long-lived electron spin in the nanosecond time regime,^[8] and so on. There is increasing interest in metallic 2D-TMDs in which a rich variety of intrinsic correlated electronic phases can be experimentally accessed in the monolayer limit.^[9-12] These electronic phases can be exploited as potential building blocks for functional van der Waals heterostructures,^[13,14] as well as for other electronic,[15,16] sensing,[17] or catalvtic^[18,19] applications.

As an example of metallic 2D-TMDs, VSe₂ can theoretically exhibit two

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polymorphs, that is, the 1T and 2H structures, but only the former is known to exist.^[20-22] The metallic character of VSe₂ stems from its 3d¹ electron configuration, which also invokes charge-density wave (CDW) order and magnetism. Recent scanning tunneling microscopy (STM) and angle-resolved photoemission (ARPES) studies have confirmed an enhancement of the CDW order in monolayer VSe₂ grown by molecularbeam epitaxy (MBE).^[23-26] However, the ferromagnetic state, predicted by first-principles calculations, remains controversial, as conflicting results have been reported.^[21,23-25,27,28] Theoretically, the magnetic moment in monolayer VSe2, mainly arising from the vanadium ions with a value of $\approx 0.6 \ \mu_{\rm B}$ per V ion,^[21,22] should be much smaller than those in 3d ferromagnets, such as Fe ($\approx 2.2 \ \mu_B$) or Co ($\approx 1.8 \ \mu_B$). Experimentally, a huge value of almost an order larger than Fe and Co has been reported, however.^[23] Such inconsistency has thus cast doubt on the sole use of bulk magnetic tools for characterizing monolayer VSe₂ and other 2D magnets, where its capability to extract a weak signal from an overwhelmingly larger substrate background signal can potentially lead to misleading results.^[29] A similar issue has been encountered previously with studies on diluted magnetic semiconductors, where unintended magnetic contributions from various sources (contaminations, defects, etc.) led to observations that were thought to be intrinsic to the materials.^[30] Therefore, to avoid any conflicting observations, careful assessments by comparison to other unambiguous techniques would be mandatory.

In this work, we utilize a combination of in situ microscopic and spectroscopic techniques, including STM/scanning tunneling spectroscopy (STS), synchrotron-based PES, ARPES, and X-ray absorption (XAS), to elucidate the metallic 1T-phase and 3d¹ electronic configuration of monolayer VSe₂ grown on HOPG by MBE. In stark contrast to previous reports,^[23,24] our element-specific X-ray magnetic circular dichroism (XMCD) measurements find no salient feature of ferromagnetically coupled V 3d states down to 16 K and up to a 1 T magnetic field. We, instead, observe traits of spin frustration in the monolayer from complementary temperature- and field-dependent

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susceptibility measurements. Our results are attributed to atomic-scale structural details of the monolayer, which can affect the relative stability of the ferromagnetic and antiferromagnetic (AF) ground states. This new finding is expected to open up new and careful searches for exotic low-dimensional quantum phenomena.

The 1T structure of VSe₂ consists of a hexagonal plane of V atoms sandwiched by two Se atomic planes with in-plane and out-of-plane lattice constants of 3.4 and 6.1 Å (Figure 1a).^[20] For its monolayer on HOPG, we find similar lattice structure and parameters by STM at 77 K. The line profile (Figure 1b) across the step edge of a VSe₂ terrace indicates an apparent height of 7.0 Å, in good agreement with the monolayer height for VSe₂. Our atomic-resolution STM measurement (Figure 1c) reveals a (1×1) hexagonal structure unit with an in-plane lattice constant of 3.4 ± 0.1 Å, as marked by the line profile in Figure 1d. We also observe two additional ($\sqrt{3} \times 2$) and ($\sqrt{3} \times \sqrt{7}$) superstructures, similarly reported in a recent STM study.^[24] These superstructures have been associated with a low-temperature CDW, with enhanced charge ordering arising from an energy gap opening in the monolayer 2D Fermi surface rather than in only the well-nested portions as in the bulk.^[25] Our STS measurements identify the presence of such a CDW gap, manifested as a differential conductance dip around the Fermi level. The size of this gap is ≈26 meV, comparable with values reported previously.^[23-25] It is worthwhile to note that the much larger CDW gap observed in a prior ARPES measurement^[26] is related to the fact that such gap was obtained along the M-K direction of the VSe₂'s 2D Brillouin zone, which has a more pronounced gap opening ($\approx 100 \text{ meV}$) than that along the Γ -K direction (in the order of ≈ 20 meV or smaller). This essentially explains the smaller CDW gap in our case, as the STS mainly probes the local density of states around the Γ point.

Figure 2 summarizes the surface electronic properties of monolayer VSe₂ measured by synchrotron-PES. Core-level measurements (Figure 2a) show that the V 2p characteristic peaks of the monolayer are positioned at 513.4 and 520.9 eV in binding energy (BE), and the Se 3d peaks at 53.4 and 54.2 eV. For atomically thin materials, one should consider the role of substrate-induced core-hole screening in defining the measured core-level positions. As a general rule, the more conductive a substrate is, the stronger the screening of the overlayer corehole. Hence, the overlayer core-levels will be shifted to lower BE. We note that this screening mechanism is rarely discussed in previous studies of 2D-TMDs,^[18,31-34] and explains the lower core-level positions of monolayer VSe2 on HOPG compared to those (V 2p_{3/2} at 514-517 eV) on insulating substrates (SiO₂ and mica). However, as we will discuss below, inevitable oxidation of ambient-exposed VSe₂ can also influence these corelevel positions.

The valence band and work function of monolayer VSe₂ are probed by a photon source with an energy of 60 eV. A sharp Fermi edge is observed at the valence band (Figure 2c), originating from the V 3d states. The spectral features at 1–6 eV are assigned to the Se 4p derived bands, while beyond 6 eV lies a substrate-related bump. From the secondary electron cutoff (Figure 2d), we obtain a work function of 5.0 ± 0.1 eV. This value, an average over the whole sample surface, has been calibrated against that of HOPG (Figure S1, Supporting www.advancedsciencenews.com





Figure 1. Crystal structure and STM/STS data of monolayer VSe₂. a) Crystal structure of 1T-VSe₂. b,c) Large-scale ($50 \times 50 \text{ nm}^2$; $V_{tip} = +1.0 \text{ V}$, $I_{set} = 200 \text{ pA}$) and atomic resolution ($5 \times 10 \text{ nm}^2$; $V_{tip} = +0.2 \text{ V}$, $I_{set} = 200 \text{ pA}$) STM images of monolayer VSe₂, measured at 77 K. The line profile in (b) shows a monolayer step height of $\approx 7 \text{ Å}$, and that in (d) reveals an in-plane lattice parameter of $3.4 \pm 0.1 \text{ Å}$. Both values are in good agreement with those in (a) for 1T-VSe₂. e) Averaged STS spectra of monolayer VSe₂ (set-point: $V_{tip} = +0.1 \text{ V}$, $I_{set} = 173 \text{ pA}$, 625 Hz, 50 mV) shows a differential conductance dip around the Fermi level with a size of $\approx 26 \text{ meV}$, due to a CDW order. Its inset shows a zoom-in of the gap feature.

Information) and is comparable with the value experimentally extracted by STM.^[35] Given its good match to that of monolayer $MoSe_2$ (5.1 eV), our measured value raises a tantalizing prospect of a low-resistivity p-type metal/semiconductor van der Waals heterostructure.^[36]

Air exposure results in newly developed peaks at the higher BE side of the initial V 2p core-levels, in addition to a high-BE shoulder in the Se 3d spectrum. These new peaks indicate oxide formation, given their similar BE positions to those of VO₂.^[37] Both the valence band and work function are significantly modified. The initially sharp Fermi edge is seen to lose almost all of its spectral weight after the exposure, accompanied by a work function reduction of 0.25 eV. These are collectively comparable to published results comparing pristine and contaminated VSe₂ crystals.^[38]

Figure 3a shows the d¹ (V⁴⁺) electronic configuration for 1T-VSe₂. Due to the octahedral coordination, the 3d degenerate orbitals of the V ions are split into two sets of e_g and t_{2g} states, with an energy separation of 10 *Dq*. The t_{2g} states are at lower energy orbitals than the e_g states and therefore are filled

by an unpaired electron. By measuring its V L_{2,3}-edge XAS in the total electron yield (TEY) geometry (Figure 3b),^[39,40] we directly verify this d¹ configuration in our monolayer. The XAS spectrum in Figure 3c reveals two main peaks (feature a,c) at \approx 518 and 524 eV, corresponding to the V L₃ and L₂ absorption edges due to dipole-allowed atomic-like transitions from the spin–orbit split V 2p_{3/2} and 2p_{1/2} core-levels to the 3d unoccupied states. The fine structures (labeled as a,b,d) at the lower energy side of those main peaks are related to the distribution of atomic multiplets. These spectral features are akin to those measured from bulk 1T-VS₂^[33] and other 3d¹ vanadium compounds,^[41] thus providing a spectroscopic fingerprint of the 1T phase of the monolayer.

In agreement with ref. [25], our V $L_{2,3}$ -edge XMCD measurements, performed in an applied magnetic field up to 1 T and at temperatures down to 16 K (lower panel of Figure 3c and Figure S2, Supporting Information), rule out intrinsic ferromagnetism in the monolayer. The measured XMCD values are less than 0.2%, far below the detection sensitivity of the synchrotron facilities used. The monolayer's SQUID *M*–*H* curves







Figure 2. Synchrotron-PES data of pristine and ambient-exposed monolayer VSe₂. a,b) V 2p and Se 3d core-levels of monolayer VSe₂. The V $2p_{3/2}$ and $2p_{1/2}$ peaks are located at 513.4 and 520.9 eV, respectively, thus indicating a spin-orbit splitting of 7.5 eV. The Se $3d_{5/2}$ and $3d_{3/2}$ peaks are positioned at 53.4 and 54.2 eV, respectively. c) The valence band of the monolayer shows a sharp Fermi edge originating from the V $3d_z^2$ band. The spectral features between 1 and 5 eV are contributed by the Se 4p derived bands, while the broad bump beyond 6 eV is a substrate peak. d) Work function of monolayer VSe₂ extracted from the secondary electron cutoff, using a photon energy of 60 eV. The upper panel (red spectra) shows the effects of air exposure. In particular, additional peaks, which are located at the higher BE side of the main peaks, are seen to develop at the V 2p and Se 3d core-levels. The sharp Fermi edge of the initial valence band has been reduced substantially, accompanied with a work function decrease of 0.25 eV.

(Figure 4a,b) support this finding and further reveal no sign of magnetic hysteresis but a linear behavior that persists down to 2 K (see Figure S3, Supporting Information, for Se-capped HOPG signal). Such a linear behavior is anisotropic in nature, with a higher moment value along the out-of-plane direction

than that within the 2D plane. The positive slope for the out-ofplane direction indicates a paramagnetic behavior, whereas the negative slope obtained within the 2D plane suggests a diamagnetic response. The latter is a primary result of the difficulty in separating the much weaker monolayer signal from the more



Figure 3. V 3d¹ electronic configuration of 1T-VSe₂, and element-specific XAS and XMCD measurements of monolayer VSe₂. a) Schematic diagram of the 3d electronic states of 1T-VSe₂ in an octahedral crystal field. The 3d orbitals of the V⁴⁺ ion are split into two sets of e_g and t_{2g} orbitals, separated by an energy $\Delta_{oct} = 10 Dq$. b) TEY detection geometry for the XAS and XMCD data shown in (c). c) Upper panel shows the XAS spectra of the monolayer measured at 30 K. Spectra highlighted in red and green are, respectively, the XAS obtained with opposite magnetic field (\approx ±300 Oe) directions, and their sum gives the total XAS marked in purple. The atomic multiplet structure, identified as a–e, provides direct proof of the d¹ configuration. The lower panel in (c) shows the XMCD signal that is within the experimental error, indicating no signs of a ferromagnetic coupling in the monolayer. This result is supported by measurements taken at 16 K and with a 1 T field (Figure S5, Supporting Information).







Figure 4. Temperature- and magnetic-field-dependent magnetic moment per area (*M*) and susceptibility ($\chi = M/H$) of monolayer VSe₂. a) In-plane and out-of-plane *M*–*H* curves measured at 300 K. b) Out-of-plane *M*–*H* curve as a function of temperature. c) Upper panel: Magnetic susceptibility χ of monolayer VSe₂. An overlap between the field-cooling (FC) and zero-field-cooling (ZFC) curves down to 2 K and in a 7 T field, together with a broad maximum for the measured χ , a characteristic feature of low-dimensional magnetic systems with a short-range antiferromagnetic (AF) interaction, indicates the presence of spin frustration in monolayer VSe₂. Lower panel: A negative Weiss constant θ_{cw} obtained from the Curie–Weiss (CW) fit to the high-temperature region of 1/ χ , suggesting an AF origin of the exchange interaction in monolayer VSe₂. d) A triangular spin-lattice with AF coupling. In this geometry, three neighboring spins cannot be mutually antialigned, thus leading to spin fluctuation and suppressed AF correlation.

predominant substrate background, a general issue known for other existing 2D magnets. $^{[23,29]}$

Magnetic susceptibility ($\chi = M/H$, where M and H are defined as magnetic moment per area and magnetic field strength, respectively) of the monolayer further reveals clear signatures of spin frustration (Figure 4c). One observes an overlap between the field-cooling (FC) and zero-field-cooling (ZFC) curves, indicating an absence of a long-range magnetic order, even down to 2 K and in a 7 T field (upper panel). One also observes a broad maximum for the measured χ , a characteristic feature of low-dimensional magnetic systems with a short-range AF interaction.^[42] A negative Weiss constant $\theta_{\rm cw}$ obtained from the Curie-Weiss (CW) fits to the high-temperature region of χ^{-1} (lower panel of Figure 4c and Figure S4, Supporting Information) indeed supports an AF origin of the exchange interaction in monolayer VSe₂. This interaction is, however, distinguished from that of a conventional AF magnet, because of its field-dependent θ_{cw} , as compared between Figure 4c and Figure S4 (Supporting Information). For the 7 T field data, we obtain a frustration factor $f = |\theta_{cw}|/T_N \ge 51$, where the AF Néel temperature T_N is below 2 K. The fact that fis much larger than unity^[43] gives further evidence of spin frustration in monolayer VSe2. The frustration observed here can have a geometrical origin, given a reasonable reproducibility of the temperature-dependent χ by the 2D spin-1/2 Heisenberg antiferromagnet on a triangular lattice model^[44,45] (Figure S5, Supporting Information). The essential idea of this model is depicted in Figure 4d, where, on a triangular spin lattice with AF coupling, three neighboring spins cannot be mutually antialigned with each other, thus resulting in fluctuating spins and suppressed AF correlation, in accordance to our observation. It is worth mentioning that, due to the small spins (i.e., spin-1/2) and monolayer nature of our VSe₂, other contributions, such as quantum frustration and low-dimensionality, may also play a role in the observed frustration, because of their tendencies to destabilize long-range magnetic order.

Given the prior predictions of a more stable ferromagnetic ground state than that of AF in monolayer VSe2,[21,22,28] our findings above may appear surprising. However, density functional theory (DFT) calculations also predict the 2H phase to be slightly more stable than the 1T phase, which in view of our ARPES measurements is incorrect (Figure 5 and Figure S6, Supporting Information). Inclusion of a Hubbard on-site repulsion U in the DFT calculations, however, stabilizes both the 1T structure, as well as the AF ordering. Nevertheless, the energy difference with ferromagnetic ordering is small (≈22 meV, Table S1, Supporting Information), which can be lifted by the presence of structural features, such as misaligned domains and boundaries, vacancies, or edges. These can contribute additional states at the Fermi level, as corroborated by ARPES. Figure 5a shows an ARPES intensity map of monolayer VSe₂ measured at 300 K. Near the Fermi level, one observes a weakly dispersive V 3d band and a set of degenerate Se 4p bands with a strong downward dispersion. These bands are better distinguished in Figure 5b,c. Although these measured bands are azimuthally averaged results of differently oriented domains (Figure S7, Supporting Information), the good agreement in the overall band structure between our experiment and DFT calculations (Figure S6, Supporting Information) confirms the corresponding 1T structure of the monolayer. Our findings are also consistent with recent reports that the intrinsic valence electron structure of monolayer VSe₂ is largely preserved on graphitic systems.^[25,26] Upon cooling to 12 K, a CDW transition is evident, shifting the V 3d and Se 4p bands away from the Fermi level, with a marked increase in their intensities (Figure 5d-f). These bands appear as a double-peak structure in the energy distribution curves (EDCs) in Figure 5g, which is otherwise unresolved at 300 K. The relative spectral changes observed are further analyzed by normalizing the EDC intensities to the Fermi-Dirac (FD) function, in order to highlight the effect of temperature on the spectral broadening. Essentially, the EDCs indicate a leading-edge midpoint shift toward







Figure 5. ARPES data of monolayer VSe₂. a) Intensity map as a function of the surface momentum component $k_{||}$ as acquired at 300 K. b) Zoom-in of (a) in the Fermi level BE range. c) Experimental band dispersion extracted from (b) using a second derivative filter. b–f) Same as (a)–(c), with data acquired at 11 K. Zero energy represents the Fermi level position. Upon cooling, the V 3d and Se 4p bands are observed to shift toward higher BE. g) These bands appear as a double-peak structure in the EDCs, which is not resolved at 300 K. The measured intensities of the EDCs have been normalized by the FD function, in order to highlight the effect of temperature on the spectral broadening. A CDW gap opening is evidenced by a leading-edge midpoint shift toward higher BE when cooling down to 11 K from 300 K.

higher BE upon cooling, indicative of the opening of a CDW gap, in agreement with our STS measurements as well as previous studies.^[23,25,26] However, we note that a full gap has been reported previously,^[25] in contrast to our observation of a finite Fermi-edge cutoff, even at 12 K. We attribute this partly to the sharp edge structures in our monolayer (Figure S7, Supporting Information), arising from the relatively high growth temperature used for this work. To support this, Figure S8 (Supporting Information) shows the density of states of a local step edge of the monolayer, probed by STS, where finite electronic states are clearly evidenced around the Fermi level. This finding is in line with the DFT calculations for both armchair and zigzag edges of monolayer VSe₂ (Figure S9, Supporting Information). These edge-derived states not only can hide the CDW gap but also induce additional energy broadening in the EDCs.

Our first experimental observation of spin frustration in monolayer VSe₂, a representative van der Waals monolayer magnet in the family of 2D-TMDs, is striking. As a frustrated magnet, its spins exhibit subtle correlations, albeit in the absence of a long-range order. Many exotic quantum phases, such as quantum spin liquid^[46] and high-temperature super-conductivity,^[47] are related to spin frustration. We, therefore, expect this work to stimulate new work in the field of metallic 2D-TMDs in the theoretical and experimental exploration of quantum phenomena in the 2D limit. Moreover, our results suggest that atomic-scale structural details of monolayer VSe₂ may play a role in its magnetism, by modifying the Fermi level

electronic structure and the relative stability of ferromagnetic and AF ground states. A systematic study establishing a structure–magnetism relationship will thus provide timely insights into this issue.

Experimental Section

Molecular-Beam Epitaxy: Monolayer VSe₂ films were grown on HOPG (SPI-1) in a custom-built MBE chamber with a base pressure of better than 1×10^{-9} mbar. The substrates were prepared by in situ cleavage followed by annealing at 820 K for at least 120 min. High-purity V and Se were evaporated from an electron-beam evaporator and a standard Knudsen cell, respectively. The Se/V flux ratio was controlled to be >10. During the growth process, the substrate temperature was kept at 650 K. To protect the samples from ambient contaminations during ex situ transport to other ultrahigh vaccum (UHV) measurement systems, a Se capping layer with a thickness of at least ~10 nm was deposited on the samples after growth. For subsequent characterization by PES, ARPES, and XAS/XMCD, the Se cap was desorbed in UHV at 500 K for 20 min.

Scanning Tunneling Microscopy and Spectroscopy: STM measurements were carried out in a custom-built multichamber UHV system housing an Omicron LT-STM interfaced with a Nanonis controller. The base pressure was better than 10^{-10} mbar. A chemically etched tungsten tip was used, and the sample was kept at 77 K during all the measurements. STM images were recorded in constant current mode. For dI/dV spectra, the tunneling current was obtained using the lock-in technique.

Synchrotron-Based Photoemission: PES measurements were performed at 300 K at the SINS beamline of the Singapore Synchrotron Light Source (SSLS), which covers the photon energy range from 50 to 1200 eV.





A Scienta SES-200 spectrometer was used to collect the spectroscopic data at normal emission, while the X-ray beam was at an incident angle of 45° relative to the sample normal. A bias voltage of -7.0 V was applied to the sample during work function measurement in order to negate the effect of the analyzer work function. The binding energy of the data was calibrated using the 4f core-levels and Fermi edge of a reference Au foil.

Angle-Resolved Photoemission: The ARPES measurements were collected with HeI α ($h\nu = 21.218$ eV) radiation source (SCIENTA VUV5k). The photoelectrons were analyzed in the plane of incidence with a high energy and angular resolution SCIENTA DA30L analyzer. The angular detection range spans $\pm 15^{\circ}$ with respect to the spectrometer lens axes. Wider angular limits were obtained by rotating the sample with respect to the analyzer lens entrance axes. During ARPES acquisition the total energy resolution was set to 20 meV, with the angular resolution being better than 0.2°. The binding energy scale was referred to the Fermi level (E_F) as measured for a clean gold substrate.

X-Ray Absorption Spectroscopy and Magnetic Circular Dichroism: XAS and XMCD measurements were independently carried out at the APE-HE beamline of Elettra Synchrotron Laboratory^[48] and at the beamline 6.3.1 of the Advanced Light Source (ALS). The spectra were collected at a sample temperature range of 16–300 K in TEY mode, in which the sample drain current was recorded as a function of the photon energy. The angle of incidence of the photon beam was set to 45° relative to the sample normal. XMCD spectra were recorded with a fixed circular polarization of the X-rays and opposite magnetic fields up to ±1 T.

Superconducting Quantum Interference Device: Temperature- and magnetic-field-dependent magnetic moments and susceptibility were characterized by Quantum Design SQUID over a temperature range of 2–300 K and fields up to 7 T.

Calculations: The electronic structure of monolayer VSe₂ was calculated within the DFT. The DFT calculations were performed using the VASP package, utilizing the projector augmented phase wave (PAW) method,^[49] and the Perdew–Burke–Ernzerhof (PBE) exchange-correlation functional.^[50] An on-site Coulomb interaction was included within the GGA+U approach using the Dudarev method,^[51] as implemented in VASP. A value U - J = 3 eV for the V 3d electrons was chosen. A separation of 15 Å between V layers in supercell was found to be sufficient to represent an isolated monolayer. A kinetic energy cutoff of 280 eV and a Γ -centered 20 × 20 × 1 k-point mesh were employed. The lattice parameters and the atomic positions were optimized until the forces on the atoms were less than 10 meV Å⁻¹.

Supporting Information

Supporting Information is available from the Wiley Online Library or from the author.

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Conflict of Interest

The authors declare no conflict of interest.

Keywords

2D materials, monolayer magnet, spin frustration, van der Waals epitaxy, vanadium diselenide

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- Q. H. Wang, K. Kalantar-Zadeh, A. Kis, J. N. Coleman, M. S. Strano, Nat. Nanotechnol. 2012, 7, 699.
- [2] M. Chhowalla, H. S. Shin, G. Eda, L. J. Li, K. P. Loh, H. Zhang, Nat. Chem. 2013, 5, 263.
- [3] D. Xiao, G. B. Liu, W. X. Feng, X. D. Xu, W. Yao, Phys. Rev. Lett. 2012, 108, 196802.
- [4] Y. Zhang, T. R. Chang, B. Zhou, Y. T. Cui, H. Yan, Z. K. Liu, F. Schmitt, J. Lee, R. Moore, Y. L. Chen, H. Lin, H. T. Jeng, S. K. Mo, Z. Hussain, A. Bansil, Z. X. Shen, *Nat. Nanotechnol.* **2014**, *9*, 111.
- [5] K. F. Mak, C. Lee, J. Hone, J. Shan, T. F. Heinz, Phys. Rev. Lett. 2010, 105, 136805.
- [6] K. F. Mak, K. L. He, J. Shan, T. F. Heinz, Nat. Nanotechnol. 2012, 7, 494.
- [7] H. L. Zeng, J. F. Dai, W. Yao, D. Xiao, X. D. Cui, Nat. Nanotechnol. 2012, 7, 490.
- [8] L. Y. Yang, N. A. Sinitsyn, W. B. Chen, J. T. Yuan, J. Zhang, J. Lou, S. A. Crooker, *Nat. Phys.* **2015**, *11*, 830.
- [9] X. X. Xi, L. Zhao, Z. F. Wang, H. Berger, L. Forro, J. Shan, K. F. Mak, Nat. Nanotechnol. 2015, 10, 765.
- [10] Y. J. Yu, F. Y. Yang, X. F. Lu, Y. J. Yan, Y. H. Cho, L. G. Ma, X. H. Niu, S. Kim, Y. W. Son, D. L. Feng, S. Y. Li, S. W. Cheong, X. H. Chen, Y. B. Zhang, *Nat. Nanotechnol.* **2015**, *10*, 270.
- [11] X. J. Zhu, Y. Q. Guo, H. Cheng, J. Dai, X. D. An, J. Y. Zhao, K. Z. Tian, S. Q. Wei, X. C. Zeng, C. Z. Wu, Y. Xie, *Nat. Commun.* 2016, 7, 11210.
- [12] L. J. Li, E. C. T. O'Farrell, K. P. Loh, G. Eda, B. Ozyilmaz, A. H. C. Neto, *Nature* **2016**, *529*, 185.
- [13] A. K. Geim, I. V. Grigorieva, *Nature* **2013**, *499*, 419.
- [14] W. Q. Liu, P. K. J. Wong, Y. B. Xu, Prog. Mater. Sci. 2019, 99, 27.
- [15] J. Feng, X. Sun, C. Z. Wu, L. L. Peng, C. W. Lin, S. L. Hu, J. L. Yang, Y. Xie, J. Am. Chem. Soc. 2011, 133, 17832.
- [16] J. H. Zhou, L. Wang, M. Y. Yang, J. H. Wu, F. J. Chen, W. J. Huang, N. Han, H. L. Ye, F. P. Zhao, Y. Y. Li, Y. G. Li, *Adv. Mater.* **2017**, *29*, 1702061.
- [17] J. Feng, L. L. Peng, C. Z. Wu, X. Sun, S. L. Hu, C. W. Lin, J. Dai, J. L. Yang, Y. Xie, Adv. Mater. 2012, 24, 1969.
- [18] M. Y. Yan, X. L. Pan, P. Y. Wang, F. Chen, L. He, G. P. Jiang, J. H. Wang, J. Z. Liu, X. Xu, X. B. Liao, J. H. Yang, L. Q. Mai, *Nano Lett.* **2017**, *17*, 4109.
- [19] J. T. Yuan, J. J. Wu, W. J. Hardy, P. Loya, M. Lou, Y. C. Yang, S. Najmaei, M. L. Jiang, F. Qin, K. Keyshar, H. Ji, W. L. Gao, J. M. Bao, J. Kono, D. Natelson, P. M. Ajayan, J. Lou, *Adv. Mater.* **2015**, *27*, 5605.
- [20] K. Tsutsumi, Phys. Rev. B 1982, 26, 5756.

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- [21] Y. D. Ma, Y. Dai, M. Guo, C. W. Niu, Y. T. Zhu, B. B. Huang, ACS Nano 2012, 6, 1695.
- [22] H. Pan, J. Phys. Chem. C 2014, 118, 13248.
- [23] M. Bonilla, S. Kolekar, Y. Ma, H. C. Diaz, V. Kalappattil, R. Das, T. Eggers, H. R. Gutierrez, P. Manh-Huong, M. Batzill, *Nat. Nanotechnol.* 2018, 13, 289.
- [24] G. Duvjir, B. K. Choi, I. Jang, S. Ulstrup, S. Kang, T. T. Ly, S. Kim, Y. H. Choi, C. Jozwiak, A. Bostwick, E. Rotenberg, J. G. Park, R. Sankar, K. S. Kim, J. Kim, Y. J. Chang, *Nano Lett.* **2018**, *18*, 5432.
- [25] J. G. Feng, D. Biswas, A. Rajan, M. D. Watson, F. Mazzola, O. J. Clark, K. Underwood, I. Markovic, M. McLaren, A. Hunter, D. M. Burn, L. B. Duffy, S. Barua, G. Balakrishnan, F. Bertran, P. Le Fevre, T. K. Kim, G. van der Laan, T. Hesjedal, P. Wahl, P. D. C. King, *Nano Lett.* **2018**, *18*, 4493.
- [26] P. Chen, W. W. Pai, Y. H. Chan, V. Madhavan, M. Y. Chou, S. K. Mo, A. V. Fedorov, T. C. Chiang, *Phys. Rev. Lett.* **2018**, *121*, 196402.
- [27] K. Xu, P. Z. Chen, X. L. Li, C. Z. Wu, Y. Q. Guo, J. Y. Zhao, X. J. Wu, Y. Xie, Angew. Chem., Int. Ed. 2013, 52, 10477.
- [28] A. Wasey, S. Chakrabarty, G. P. Das, J. Appl. Phys. 2015, 117, 064313.
- [29] D. J. O'Hara, T. Zhu, A. H. Trout, A. S. Ahmed, Y. K. Luo, C. H. Lee, M. R. Brenner, S. Rajan, J. A. Gupta, D. W. McComb, R. K. Kawakami, *Nano Lett.* **2018**, *18*, 3125.
- [30] T. Dietl, Nat. Mater. 2010, 9, 965.
- [31] Z. P. Zhang, J. J. Niu, P. F. Yang, Y. Gong, Q. Q. Ji, J. P. Shi, Q. Y. Fang, S. L. Jiang, H. Li, X. B. Zhou, L. Gu, X. S. Wu, Y. F. Zhang, Adv. Mater. 2017, 29, 1702359.
- [32] Y. Wang, Z. Sofer, J. Luxa, M. Pumera, Adv. Mater. Interfaces 2016, 3, 1600433.
- [33] M. Mulazzi, A. Chainani, N. Katayama, R. Eguchi, M. Matsunami, H. Ohashi, Y. Senba, M. Nohara, M. Uchida, H. Takagi, S. Shin, *Phys. Rev. B* **2010**, *82*, 075130.
- [34] D. Q. Gao, Q. X. Xue, X. Z. Mao, W. X. Wang, Q. Xu, D. S. Xue, J. Mater. Chem. C 2013, 1, 5909.
- [35] Z. L. Liu, X. Wu, Y. Shao, J. Qi, Y. Cao, L. Huang, C. Liu, J. O. Wang, Q. Zheng, Z. L. Zhu, K. Ibrahim, Y. L. Wang, H. J. Gao, *Sci. Bull.* 2018, *63*, 419.

- [36] M. Farmanbar, G. Brocks, Adv. Electron. Mater. 2016, 2, 1500405.
- [37] G. Silversmit, D. Depla, H. Poelman, G. B. Marin, R. De Gryse, J. Electron Spectrosc. Relat. Phenom. 2004, 135, 167.
- [38] R. Claessen, I. Schafer, M. Skibowski, J. Phys.: Condens. Matter 1990, 2, 10045.
- [39] P. K. J. Wong, E. van Geijn, W. Zhang, A. A. Starikov, T. L. A. Tran, J. G. M. Sanderink, M. H. Siekman, G. Brocks, P. J. Kelly, W. G. van der Wiel, M. P. de Jong, *Adv. Funct. Mater.* **2013**, *23*, 4933.
- [40] P. K. J. Wong, W. Zhang, K. Wang, G. van der Laan, Y. B. Xu, W. G. van der Wiel, M. P. de Jong, J. Mater. Chem. C 2013, 1, 1197.
- [41] N. B. Aetukuri, A. X. Gray, M. Drouard, M. Cossale, L. Gao, A. H. Reid, R. Kukreja, H. Ohldag, C. A. Jenkins, E. Arenholz, K. P. Roche, H. A. Durr, M. G. Samant, S. S. P. Parkin, *Nat. Phys.* 2013, 9, 661.
- [42] Y. Zhou, K. Kanoda, T.-K. Ng, Rev. Mod. Phys. 2017, 89, 025003.
- [43] L. Balents, Nature 2010, 464, 199.
- [44] P. Fazekas, P. W. Anderson, Philos. Mag. 1974, 30, 423.
- [45] N. Elstner, R. R. P. Singh, A. P. Young, Phys. Rev. Lett. 1993, 71, 1629.
- [46] L. Clark, G. Sala, D. D. Maharaj, M. B. Stone, K. S. Knight, M. T. F. Telling, X. Wang, X. Xu, J. Kim, Y. Li, S.-W. Cheong, B. D. Gaulin, *Nat. Phys.* **2019**, *15*, 262.
- [47] P. A. Lee, Rep. Prog. Phys. 2008, 71, 012501.
- [48] G. Panaccione, I. Vobornik, J. Fujii, D. Krizmancic, E. Annese, L. Giovanelli, F. Maccherozzi, F. Salvador, A. De Luisa, D. Benedetti, A. Gruden, P. Bertoch, F. Polack, D. Cocco, G. Sostero, B. Diviacco, M. Hochstrasser, U. Maier, D. Pescia, C. H. Back, T. Greber, J. Osterwalder, M. Galaktionov, M. Sancrotti, G. Rossi, *Rev. Sci. Instrum.* **2009**, *80*, 043105.
- [49] G. Kresse, J. Furthmuller, Comput. Mater. Sci. 1996, 6, 15.
- [50] J. P. Perdew, K. Burke, M. Ernzerhof, Phys. Rev. Lett. 1996, 77, 3865.
- [51] S. L. Dudarev, G. A. Botton, S. Y. Savrasov, C. J. Humphreys, A. P. Sutton, Phys. Rev. B 1998, 57, 1505.

